

## H<sub>2</sub>S and CH<sub>3</sub>SH Sensor Using a Thick Film of Gold-Loaded Tungsten Oxide

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A thick film sensor element fabricated with 0.5 wt% Au-loaded WO<sub>3</sub> was found to exhibit excellent sensing properties to H<sub>2</sub>S and CH<sub>3</sub>SH in air at 300 °C. The element was sensitive to very dilute H<sub>2</sub>S and CH<sub>3</sub>SH, the detection limits of the gases being 20 ppb and 30 ppb, respectively.

Gas sensors using semiconducting oxides are now used extensively for gas leakage alarms for usual inflammable gases such as H<sub>2</sub>, CH<sub>4</sub> and C<sub>3</sub>H<sub>8</sub>. Recently much attention is being paid on the development of those semiconductor sensors that detect various smells or, more precisely, typical constituents of smells. For example, sensor elements sensitive to trimethylamine<sup>1)</sup> and CH<sub>3</sub>SH<sup>2)</sup> have been investigated for the purposes of checking the freshness of fishes and a breath odor, respectively. Detection of odors is also important for our civil life. Volatile sulfides such as H<sub>2</sub>S and CH<sub>3</sub>SH are typical odorous components in living environments, which produce a sick odor even at a very low concentration, and their detection is requested for realizing an auto-ventilation system. For this purpose, the relevant sensor should be at least as sensitive as human noses, to which the threshold concentrations of H<sub>2</sub>S and CH<sub>3</sub>SH are as low as a few tens ppb. So far attempts have been made for H<sub>2</sub>S-sensitive elements,<sup>3-7)</sup> but none seems to be able to detect such dilute H<sub>2</sub>S satisfactorily. We have found that a thin film SnO<sub>2</sub> element fabricated by using hydrothermally synthesized SnO<sub>2</sub> sol shows fairly good sensing properties to dilute H<sub>2</sub>S.<sup>8)</sup> Subsequently it has been found that a thick film element using Au-loaded WO<sub>3</sub> shows far better properties, being able to detect H<sub>2</sub>S and CH<sub>3</sub>SH of 20 - 30 ppb levels satisfactorily, as described below.

The powder of WO<sub>3</sub> was prepared by pyrolyzing ammonium paratungstate ((NH<sub>4</sub>)<sub>10</sub>W<sub>12</sub>O<sub>41</sub>·5H<sub>2</sub>O) at 600 °C for 5 h in air, followed by milling the product in a zirconia-ball mill for 1 day. The powder was loaded with Au by mixing it with an aqueous dispersion of colloidal Au particles (colloid concentration : 0.5 × 10<sup>-3</sup> mol·dm<sup>-3</sup>) for 5 h under agitation, followed by filtration, washing with deionized water, drying at 110 °C, and calcination at 600 °C for 5 h. Each powder sample was mixed with an organic binder (a mixture of ethyl cellulose and α-terpineol), and the resulting paste was applied on an alumina substrate (5.0 mm x 6.0 mm), on which comb type Au electrodes were printed, and calcined at 700 °C for 4 h. The sensor films thus obtained were about 20 μm thick. Each element was put inside a conventional flow cell for exposure to the flow (500 cm<sup>3</sup>/min) of air or H<sub>2</sub>S (or CH<sub>3</sub>SH)-air mixtures. Both of air and the gas mixtures were humidified to 60% RH at 25 °C. The electrical resistance of the element was measured under the applied voltage of dc 5V. From the resistance values in air (R<sub>a</sub>) and in the gas mixtures (R<sub>g</sub>), the gas sensitivity was obtained as R<sub>a</sub>/R<sub>g</sub> as usual. The sensor temperature was

controlled by means of a Pt heater, printed on the back of the alumina substrate.

The influences of Au loadings on the H<sub>2</sub>S sensing characteristics of WO<sub>3</sub>-based elements were first investigated. Figure 1 shows the gas sensitivity to 1 ppm H<sub>2</sub>S at 300 °C as a function of Au loading. The sensitivity, starting from a small value of about 2 for the pure WO<sub>3</sub> element, increased with an increase in Au loading, and reached a maximum value of 11 at 0.5 wt% Au, whereas it decreased to lower values at 0.8 and 1.0 % Au. From these results, an optimum Au loading is judged to be 0.5 wt%. The corresponding element will be abbreviated as Au(0.5 wt%)-WO<sub>3</sub> hereafter.

Figure 2 shows the gas sensitivity of Au(0.5 wt%)-WO<sub>3</sub> element to 1 ppm H<sub>2</sub>S as a function of operating temperature. With a rise in temperature, the sensitivity increased up to 300 °C, and exhibited a maximum value of 11 at 300 °C. Then it showed a plateau value of *ca.* 10 at 350 and 400 °C before going down sharply to no more than 3 at 450 °C and above. These results indicate an optimum operating temperature of 300 °C.

The gas sensitivity of the same element to varying concentrations of H<sub>2</sub>S was examined at 300 °C. Figure 3 shows the resulting data together with those of the pure WO<sub>3</sub> element. It is noteworthy that Au(0.5 wt%)-WO<sub>3</sub> element can respond to as dilute as 10 ppb H<sub>2</sub>S in air with a sensitivity value of 1.7, giving an almost linear relationship between the logarithm of the gas sensitivity and that of the H<sub>2</sub>S concentrations of the range 10 - 1000 ppb. In contrast, the pure WO<sub>3</sub> element is seen to be far less sensitive to H<sub>2</sub>S, responding to H<sub>2</sub>S at a level of 500 ppb or above. For a practical semiconductor gas sensor combined with ordinary circuitry, the lower detection limit is often assumed to be the gas concentration to which the sensor exhibits a sensitivity value of 2. According to this criterion, the limiting H<sub>2</sub>S concentration for the Au(0.5 wt%)-WO<sub>3</sub> element is estimated to be 20 ppb, to be compared with the corresponding value of 1000 ppb for the pure WO<sub>3</sub> element.

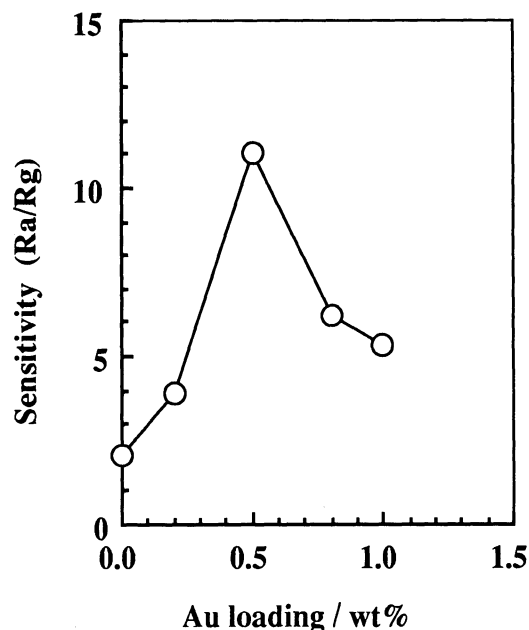


Fig.1. Sensitivity of Au-loaded WO<sub>3</sub> elements to 1 ppm H<sub>2</sub>S at 300 °C as a function of Au loading.

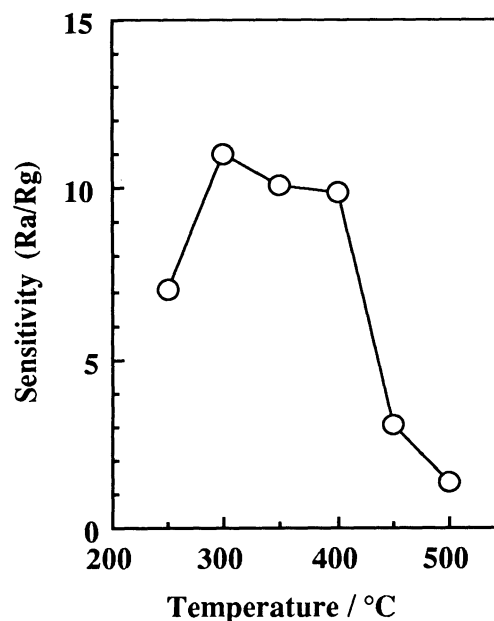


Fig.2. Sensitivity of 0.5 wt% Au-loaded WO<sub>3</sub> element to 1 ppm H<sub>2</sub>S at various operating temperatures.

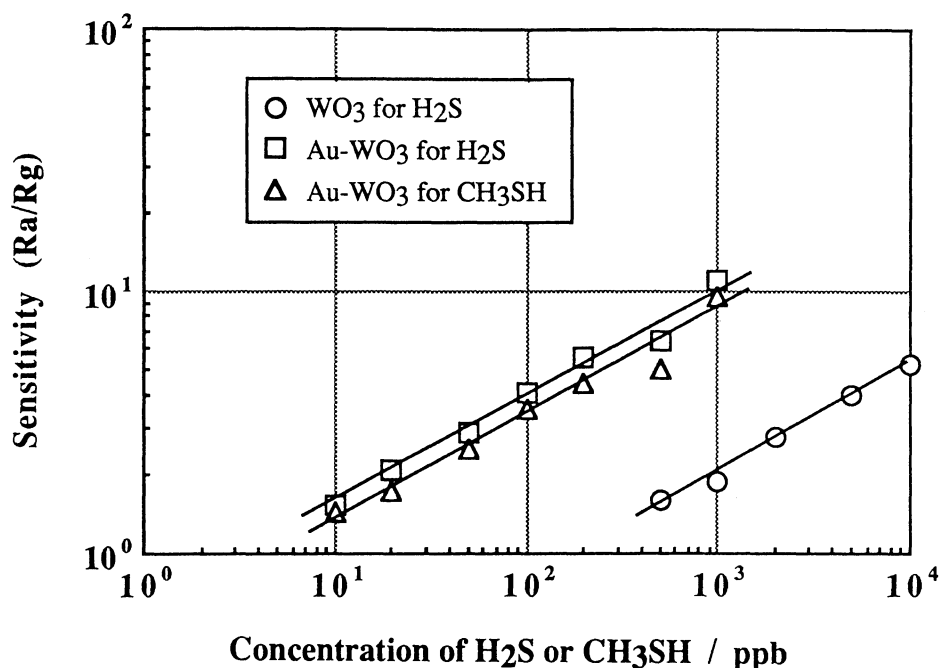


Fig.3. Sensitivity to H<sub>2</sub>S or CH<sub>3</sub>SH for the sensor element using pure or 0.5 wt% Au-loaded WO<sub>3</sub> as a function of gas concentration (300 °C).

Figure 4 shows the response transient of Au(0.5 wt%)-WO<sub>3</sub> element to 100 ppb H<sub>2</sub>S in air at 300 °C. On turning-on H<sub>2</sub>S the sensor resistance decreased rapidly, reaching a steady state in about 2 min, and on turning-off it recovered the initial value in about 3 min. The response rate became somewhat slower as the concentration of H<sub>2</sub>S decreased, but the 90% response time to 20 ppb H<sub>2</sub>S was still 2.5 min. This assures that the present element has practically acceptable transient characteristics to dilute H<sub>2</sub>S.

It was further found that the Au(0.5 wt%)-WO<sub>3</sub> element was also very sensitive to CH<sub>3</sub>SH at 300 °C. As seen from the gas sensitivity vs. CH<sub>3</sub>SH concentration correlation over a range of 10 - 1000 ppb indicated in Fig. 3, the detection limit of the element was again as low as 30 ppb CH<sub>3</sub>SH. The response rates to CH<sub>3</sub>SH were much the same as those to H<sub>2</sub>S.

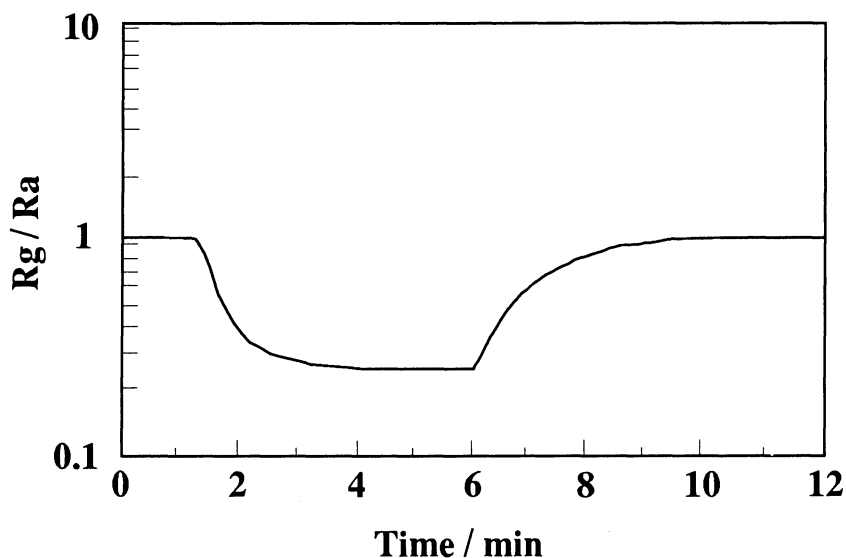


Fig.4. Response transient of 0.5 wt% Au-loaded WO<sub>3</sub> element to 100 ppb H<sub>2</sub>S at 300 °C.

As mentioned above, a thick film element using  $\text{WO}_3$  becomes very sensitive to  $\text{H}_2\text{S}$  and  $\text{CH}_3\text{SH}$  when loaded with Au. In the case of a thin film element using  $\text{SnO}_2$  previously reported,<sup>8)</sup> rather high sensitivity to  $\text{H}_2\text{S}$  is obtained without the addition of such a promoter. It appears that  $\text{WO}_3$  is less reactive to  $\text{H}_2\text{S}$  than  $\text{SnO}_2$  so that it needs an assistance of a weak promoter like Au. We have reported that a sintered block type element using Au- $\text{WO}_3$  is highly sensitive to  $\text{NH}_3$  and that the high sensitivity benefits from an electronic interaction between Au and  $\text{WO}_3$ <sup>9)</sup>; The interaction presumably takes place through the formation of adsorbed oxygen at the Au/ $\text{WO}_3$  interface and decreases the conduction electrons of  $\text{WO}_3$ . A similar mechanism can be operative in the present case. That is, the electrical resistance of the element decreases from a very high level in air to a level comparable to that of a pure  $\text{WO}_3$  element when the adsorbed oxygen at the interface is consumed upon exposure to  $\text{H}_2\text{S}$ , giving rise to the high gas sensitivity of the Au-loaded  $\text{WO}_3$  element. It should be pointed out, however, that a structural factor is also involved in the high  $\text{H}_2\text{S}$  sensitivity of the present thick film element, since the block type element using the same material has been confirmed to be hardly sensitive to  $\text{H}_2\text{S}$  despite its high sensitivity to  $\text{NH}_3$  as mentioned. This seems to be ascribed to the differences in gas diffusivity and reactivity between  $\text{H}_2\text{S}$  and  $\text{NH}_3$ . Assuming Knudsen diffusion through the pores of the element, the rates of gas diffusion are in the order  $\text{H}_2\text{S} \approx \text{O}_2 < \text{NH}_3$ . It has been pointed out by Egashira *et al.*<sup>10)</sup> that the thickness of the gas sensing layer affects the gas sensitivity differently depending on the diffusivity of the gas in problem relative to that of  $\text{O}_2$ . It is considered that the sensitivity to  $\text{H}_2\text{S}$  (or  $\text{CH}_3\text{SH}$ ) becomes more favorable with a thinner sensing layer, while the reverse is true for that to  $\text{NH}_3$ . In addition,  $\text{H}_2\text{S}$  appears to be more reactive on the oxide surface than  $\text{NH}_3$ , making the thinner sensing layer even more favorable for the  $\text{H}_2\text{S}$  detection. These would explain why the high sensitivity to  $\text{H}_2\text{S}$  (or  $\text{CH}_3\text{SH}$ ) was attained only with the present thick film element using Au- $\text{WO}_3$ .

In conclusion, the thick film using Au-loaded  $\text{WO}_3$  exhibited excellent sensing properties to  $\text{H}_2\text{S}$  and  $\text{CH}_3\text{SH}$  in air at 300 °C. The detection limit of the film was as low as 20 ppb for  $\text{H}_2\text{S}$  and 30 ppb for  $\text{CH}_3\text{SH}$ .

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